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Hydrogen processing as a way of producing fine metallic powders

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Hydrogen (H) commonly undergoes reactions with metallic elements forming metal hydrides (M-H) since metals can exhibit hydrogen diffusivity comparable to that of ions in aqueous solutions, allowing thermodynamic equilibrium to be attained relatively quickly, even at room temperature [1]. The capacity of hydrogen to form metal-hydride systems is generally attributed to its (a) moderate electronegativity, (b) small atomic size, and (c) low nuclear mass [2]. Metal lattices accommodate H atoms typically at their interstitial sites by going through fundamental changes in the crystal structure leading to the formation of different phases metal-hydrogen systems [2]. H atoms can be densely packed in metal hosts through a large exothermic reaction such that the density of hydrogen can surpass that of liquid hydrogen. The aforementioned factors stimulate the utilization of metal hydrides for energy storage purposes [3].

The impact of H on metals can be examined from various perspectives. Hydrogen embrittlement (HE) is a detrimental phenomenon for structural metals, as it can proceed to sudden failure even at low stress levels and concentrations of H. Conversely, the embrittlement is regarded as a convenient method for producing finely divided metal powders of transition and rareearth (RE) metals which exhibit brittleness after H absorption. The brittle nature of H absorbing metallic systems stems from the significant volumetric changes that occurs during the formation of solid solutions and/or hydrides. Intermetallics such as LaNi5, SmFe3 and TiFe display high H absorption capacity and exceptional lattice expansion around 27.4%, 19%, and 18.8% respectively which results in the decrepitation of the material into fine powder form readily [4]. Hydrogen decrepitation (HD) is also a widely used route to obtain sinterable powders for strong permanent magnet production (e.g., Sm₂(Co,Fe)₁₇, Nd₂Fe₁₄B). Nd-Fe-B alloys become H hardened due to the formation of a Ndrich hydride phase and disintegrate upon milling methods [5].

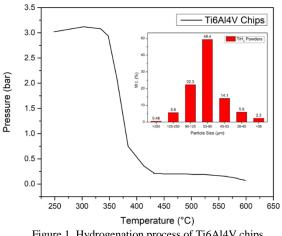


Figure 1. Hydrogenation process of Ti6Al4V chips

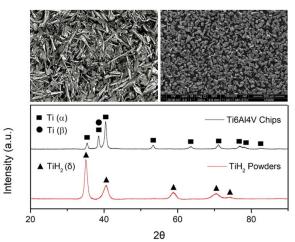


Figure 2. XRD spectra of hydrogenated Ti6Al4V powders

The elemental metals such as Pd, Ti, Zr also accomodate large amount of H in their lattice sites without losing their integrity. Therefore, most M-H systems require further mechanical force to form powders similarly to magnetic compounds.

In this study, Ti-H system is investigated by using a titanium alloy. Ti6Al4V chips are cleaned and hydrogenated at 650°C for 1.5 hours in a stainless-steel vessel. Then, the vessel is air cooled overnight for the formation of brittle TiH₂ phase. After, the chips are ball-milled for 30 minutes for the fragmentation process. Last, the powders are sieved and characterized. According to Figure 1, the hydrogenation started around 340°C and continued until the chips are fully hydrogenated. 50% of the powders were in between 53-90 µm and there were also particles below 38 µm. XRD spectra of the powders shows that there is only TiH2 phase present without any α or β phases (Fig.2.).

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